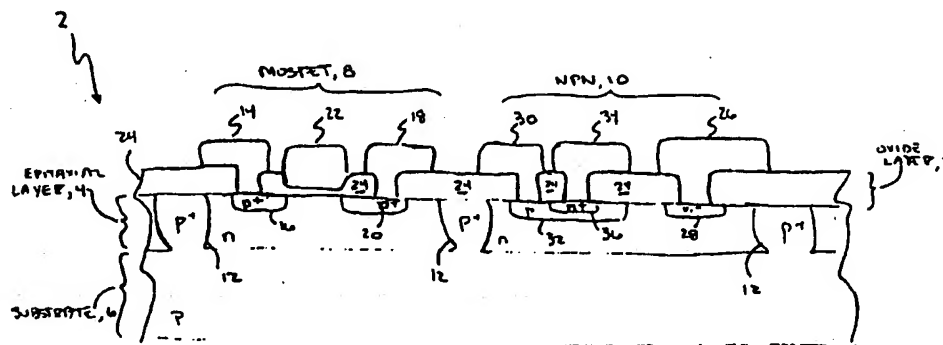




## INTERNATIONAL APPLICATION PUBLISHED UNDER THE PATENT COOPERATION TREATY (PCT)

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<b>(21) International Application Number:</b> PCT/US91/06424 <b>(22) International Filing Date:</b> 5 September 1991 (05.09.91)  <b>(30) Priority data:</b> 577,526                      5 September 1990 (05.09.90)    US 615,425                      16 November 1990 (16.11.90)    US  <b>(71) Applicant:</b> YALE UNIVERSITY [US/US]; 246 Church Street, New Haven, CT 06510 (US). <b>(72) Inventor:</b> MA, Tso-Ping ; 169 Northford Road, Branford, CT 06405 (US).  <b>(74) Agent:</b> PRAHL, Eric, L.; Fish & Richardson, 225 Franklin Street, Suite 3100, Boston, MA 02110-2804 (US).		<b>(81) Designated States:</b> AT (European patent), AU, BE (European patent), CA, CH (European patent), DE (European patent), DK (European patent), ES (European patent), FR (European patent), GB (European patent), GR (European patent), IT (European patent), JP, KR, LU (European patent), NL (European patent), SE (European patent), SU <sup>+</sup> .  <b>Published</b> <i>With international search report.          Before the expiration of the time limit for amending the claims and to be republished in the event of the receipt of amendments.</i>

**(54) Title:** ISOTOPICALLY ENRICHED SEMICONDUCTOR DEVICES**(57) Abstract**

A semiconductor structure including a single-crystal region (6, 4) composed of an isotopically enriched semiconductor material, and a semiconductor device (8, 10) formed in the isotopically enriched semiconductor region.

# + DESIGNATIONS OF "SU"

Any designation of "SU" has effect in the Russian Federation. It is not yet known whether any such designation has effect in other States of the former Soviet Union.

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ISOTOPICALLY ENRICHED SEMICONDUCTOR DEVICESBackground of the Invention

This is a continuation of application Serial No. 07/577,526, filed September 5, 1990.

5       The invention relates to semiconductor devices fabricated from semiconductor materials, such as silicon.

      Silicon, an abundantly available element in nature, has become one of the most commonly used materials for fabricating electrical and electro-optical  
10    devices. In its naturally occurring form, silicon is primarily composed of three isotopes of silicon, namely, 92.2%  $^{28}\text{Si}$ , 4.7%  $^{29}\text{Si}$ , and 3.1%  $^{30}\text{Si}$ , which is also roughly the composition of the silicon crystals used by the silicon device industry. Due to its many desirable  
15    properties, in comparison to other semiconductors, not the least of which is the ability to easily grow  $\text{SiO}_2$  insulation layers on it, silicon has provided the foundation upon which a semiconductor industry has been built.

20       Devices fabricated on single-crystal silicon have performance characteristics that are governed by the electrical and physical properties of the silicon itself. Some of the important properties of the single-crystal silicon which affect device characteristics are carrier  
25    mobilities, energy band gap, effective mass of the carriers, and thermal conductivity. Carrier mobilities, for example, govern signal transit times and thus place a limit on device speed. Thermal conductivity, on the other hand, governs power dissipation which, in turn,  
30    places an upper limit on the packing densities achievable for devices on a chip or the amount of power that can safely be generated in the circuit without significantly degrading circuit performance.

      As engineers push the performance of the silicon-  
35    based devices, the electrical and physical properties of

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the silicon often place serious limits on what is ultimately achievable from such devices. For example, as signals and data communications advanced into the higher reaches of the frequency domain, the speed limitations inherent in silicon-based devices caused engineers to shift from silicon to other semiconductor materials, such as GaAs, which exhibit substantially higher carrier mobilities and thus are capable of performance at higher frequencies. The electron mobility in GaAs is over five times greater than that the carrier mobilities typically associated with silicon. This shift to GaAs was felt necessary in spite of the greater technological difficulties associated with fabricating devices from GaAs as compared to silicon. For example, it is appreciably more difficult to grow single-crystal GaAs and to form useful insulating layers on the GaAs.

#### Summary of the Invention

In general, in one aspect, the invention is a semiconductor structure including a single-crystal region composed of an isotopically enriched semiconductor material, and a semiconductor device formed in the isotopically enriched semiconductor region.

Preferred embodiments include the following features. The semiconductor structure also includes a substrate and the isotopically enriched semiconductor region is formed as a layer on one surface of the substrate. The semiconductor material is selected from among the group consisting of silicon, germanium and GaAs. In the case of silicon, the silicon region has a higher proportion of one of the isotopes of Si than is present in naturally occurring silicon (e.g., it is composed of at least 98 %  $^{28}\text{Si}$ ). In the case of germanium, the germanium region has a higher proportion of one of the isotopes of Ge than is present in naturally

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occurring germanium (e.g., it is composed of at least 98 %  $^{74}\text{Ge}$ ). In the case of GaAs, the gallium in the GaAs region has a higher proportion of one of the isotopes of Ga than is present in naturally occurring gallium (e.g., it is composed of at least 98 %  $^{69}\text{Ga}$ ).

Also in preferred embodiments, the semiconductor device is either an electro-optical device or an opto-electronic device. The semiconductor structure further includes an integrated circuit fabricated in the isotopically enriched semiconductor region. The isotopically enriched semiconductor region primarily serves to provide heat dissipation for the device.

In general, in another aspect, the invention is a method for fabricating a semiconductor structure including the steps of forming a single-crystal region composed of isotopically enriched semiconductor material, and fabricating a semiconductor device in the isotopically enriched region.

Preferred embodiments include the following features. The isotopically enriched semiconductor region is formed as a layer on one surface of a substrate by means of epitaxial deposition. The semiconductor material is selected from a group consisting of silicon, germanium and GaAs. In the case of silicon, the silicon region has a higher proportion of one of the isotopes of Si than is present in naturally occurring silicon (e.g., it is composed of at least 98 %  $^{28}\text{Si}$ ). In the case of germanium, the germanium region has a higher proportion of one of the isotopes of Ge than is present in naturally occurring germanium (e.g., it is composed of at least 98 %  $^{74}\text{Ge}$ ). In the case of GaAs, the gallium in the GaAs region has a higher proportion of one of the isotopes of Ga than is present in naturally occurring gallium (e.g., it is composed of at least 98 %  $^{69}\text{Ga}$ ).

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In yet another aspect, the invention is a semiconductor structure including an isotopically enriched single-crystal silicon region composed of at least 98%  $^{28}\text{Si}$ . The semiconductor structure also includes  
5 a substrate and the isotopically enriched silicon region is formed on one surface of the substrate.

In still another aspect, the invention is a semiconductor structure including an isotopically enriched single-crystal GaAs region in which the gallium  
10 is composed of at least 98%  $^{69}\text{Ga}$ . The semiconductor structure also includes a substrate and the isotopically enriched GaAs region is formed on one surface of the substrate.

Isotopically enriched single-crystal silicon will  
15 exhibit significantly higher carrier mobilities and thermal conductivity than single-crystal silicon made from the sources of naturally occurring silicon. Higher carrier mobilities mean that devices fabricated from the isotopically enriched silicon will exhibit faster device  
20 speeds and higher frequency performance than was previously possible using conventional compositions of silicon. Higher carrier mobilities also imply lower resistivity and thus lower heat generation within the material. In addition, higher thermal conductivity means  
25 that the isotopically enriched silicon devices will exhibit better heat dissipation, thereby making it possible to increase device packing densities within integrated circuit chips and to increase power output per unit area of power devices. Furthermore, the invention  
30 has applicability in device structures, such as, semiconductor laser arrays, which utilize silicon substrates or silicon layers primarily for heat dissipation. Using the invention in those cases will improve the thermal performance of the devices and also  
35 make higher packing densities possible.

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The same principles and advantages also apply to other semiconductor materials such as Ge and GaAs. That is, isotopically enriched versions of those materials will exhibit higher carrier mobilities and higher thermal conductivities. As in the case of isotopically enriched silicon, using such isotopically enriched semiconductors for device fabrication will produce devices which exhibit improved performance in ways which relate to the higher carrier mobilities and higher thermal conductivities.

Other advantages and features will become apparent from the following description of the preferred embodiment and from the claims.

#### Description of the Preferred Embodiment

Fig. 1 is a cross-section of a representational portion of wafer on which an integrated circuit is fabricated; and

Fig. 2 is a schematic representation of a molecular beam epitaxy system for depositing a layer of isotopically enriched silicon;

Fig. 3 is a double-heterostructure laser made from isotopically enriched GaAs; and

Fig. 4 is a diode made from isotopically enriched Ge.

#### Structure

Referring to Fig. 1, integrated circuit 2 is fabricated in an n-type epitaxial layer 4 that is formed on one side of a p-type silicon substrate 6. N-type epitaxial layer 4 is a single-crystal silicon layer composed of isotopically enriched silicon in which the percentage of  $^{28}\text{Si}$  is greater than the percentage of that isotope typically found in naturally occurring sources of silicon. More specifically, the percentage of  $^{28}\text{Si}$  is greater than at least 94% and more preferably is greater than 98%, with the aggregate of other isotopes of

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silicon, namely,  $^{29}\text{Si}$  and  $^{30}\text{Si}$ , being reduced accordingly. In addition, in this application, a material is defined to be single-crystal if it has single-crystal structure over a distance that is larger than the dimensions of the devices that are or will be fabricated using that silicon material.

The illustrated portion of integrated circuit 2 includes a metal-oxide-semiconductor field effect transistor (MOSFET) 8 and an NPN bipolar transistor 10, each of which is isolated from the other and from other devices on the chip by  $p^+$ -type regions 12 that are diffused through epitaxial layer 4 into substrate 6. MOSFET 8 is a conventionally designed device including a drain metallization 14 in contact with a  $p^+$ -type drain region 16, a source metallization 18 in contact with a  $p^+$ -type source region 20 and a gate metallization 22 isolated from electrical contact with epitaxial layer 4 by a  $\text{SiO}_2$  layer 24.

NPN transistor 10 is also of conventional design and includes a collector metallization 26 in contact with an  $n^+$ -type region 28, a base metallization 30 in contact with a  $p$ -type base region 32 and an emitter metallization 34 in contact with an  $n^+$ -type emitter region 36.

With the exception of epitaxial layer 4, integrated circuit 2, including MOSFET 8 and NPN 10, may be produced by any of the device fabrication techniques familiar to those skilled in the art and with any of the materials commonly used to fabricate such circuits. These techniques include, for example, diffusion, ion implantation, and other well known particle beam techniques. For further discussion of such techniques, see, for example, C.W. Pearce in Chapter 2 of VLSI Technology, edited by S.M. Sze and published by McGraw-Hill, 1983; W.R. Runyan and K.E. Bean in Chapter 7 of Semiconductor Integrated Circuit Processing Technology,



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published by Addison-Wesley, 1990; and J.W. Mayor and S.S. Lau in Chapter 8 of Electronic Materials Science for Integrated Circuits in Si and GaAs, published by MacMillian, 1990.

5           Except for the fact that an isotopically enriched silicon source is used, epitaxial layer 4 may also be formed by using any of the epitaxial techniques known to those skilled in the art, including, for example, chemical vapor deposition (CVD), high vacuum CVD,  
10 molecular beam epitaxy (MBE), implantation/recrystallization, and RF sputtering. Unless isotope enrichment is built into the epitaxial process, the percentage of  $^{28}\text{Si}$  in the silicon source material should, of course, be equal to whatever  
15 proportion of that isotope is desired in epitaxial layer 4. Thus, to form an epitaxial layer having 98%  $^{28}\text{Si}$ , one should use a source in which  $^{28}\text{Si}$  is present in that proportion. Such isotopically enriched silicon sources can be obtained from commercial suppliers such as, for  
20 example, Oak Ridge National Laboratories in Oak Ridge, Tennessee or Atomergic Chemetals in Farmingdale, New York.

          The appropriate form of the source material, i.e., whether it is powdered, gaseous or liquid, depends upon  
25 the technique being used to grow the epitaxial layer. Thus, for example, if MBE is used, a powdered or solid bulk source of isotopically enriched silicon is appropriate. Whereas, if a CVD process is used, a  
gaseous source such as, for example, silane, disilane or  
30 silicon tetrachloride, would be more appropriate. If the isotopically enriched silicon is obtained in powdered or solid form, any of the other forms can readily be produced using known techniques.

          Fig. 2 illustrates the basic components of an MBE  
35 system for growing an epitaxial layer of isotopically

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enriched silicon on a silicon substrate 50. The system includes a chamber 52 connected to a vacuum pump 54 which establishes the required vacuum in chamber 52. Inside chamber 52, a support 56 holds silicon substrate 50 that is heated by a heating element 58 to an appropriate temperature for MBE. An electron beam gun 60 generates an e-beam 62 that heats up and vaporizes a source 64 of isotopically enriched silicon 60, thereby generating a molecular beam 66 of silicon that deposits an epitaxial layer on substrate 50. A dopant effusion cell 68 may be used to introduce a dopant into the epitaxial layer as it is being grown. The supply of dopant from effusion cell 68 is controlled by a mechanical shutter 72.

The deposition process is continued until an epitaxial layer of desired thickness is achieved, e.g. 2-3 microns for power devices or sub-micron thicknesses for high frequency devices. The operating parameters during deposition such as chamber pressure, the temperature of substrate 50, the temperature of effusion cell 68, the dopant source material used in effusion cell 68 and the duration of the growth cycle may be the same as those which are commonly used to deposit epitaxial layers of ordinary silicon and are well known to those skilled in the art.

During the deposition of the isotopically enriched silicon layer, precautions must be taken to assure that isotopes of silicon other than  $^{28}\text{Si}$  are not introduced into the epitaxial layer. For example, silicon from quartz tubes or containers used in the deposition system could be a source of such contamination if they are permitted to get too hot.

If the epitaxial layer is grown by ion implantation, it may not be necessary to use a source of isotopically enriched silicon. A mass separator could be used in the ion implantation equipment to select the  $^{28}\text{Si}$

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from the ion stream and to exclude the other isotopes of silicon. Ion implantation techniques, however, typically generate amorphous layers which must be recrystallized by using a high temperature anneal. Since the

5    recrystallization may not eliminate all of the defects in the epitaxial layer, the benefits of using an isotopically enriched layer (i.e. the improved carrier mobilities and the thermal conductivities) could be masked by these residual defects.

10        Other embodiments are within the following claims. For example, it should be understood that the above example of an integrated circuit was merely illustrative. Although only a MOSFET and an NPN transistor were shown, the invention described herein has applicability to all

15    electronic devices made using single-crystal silicon for which improved carrier mobility and/or improved thermal conductivity would provide a desired benefit. In this case, electronic devices is meant to include both passive devices (e.g., resistors) and active devices (e.g.,

20    transistors). The invention also has applicability to particular categories of semiconductor devices such as electro-optical devices including, for example, deflectors and modulators, and opto-electronic devices including, for example, lasers. In addition, rather

25    than using epitaxial layers made of isotopically enriched silicon, wafers made entirely of isotopically enriched silicon may be used. Such wafers may be cut from single-crystal silicon grown from a source of isotopically enriched silicon by any of the known crystal growing

30    techniques, including, for example, the commonly used Czochralski technique.

In addition, isotopically enriched silicon can be used to replace materials currently used to provide heat sinking in some semiconductor devices such as, for

35    example, semiconductor laser arrays.

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Other embodiments use other isotopically enriched semiconductor materials, such as, Ge, Ge-Si, and GaAs. As with isotopically enriched silicon, such other isotopically enriched semiconductors will exhibit improved carrier mobilities and thermal conductivity. If such materials are used for device fabrication, the resulting devices will exhibit improvements in device performance corresponding to the higher carrier mobilities and thermal conductivities.

In its naturally occurring form, germanium is composed of about 7.8%  $^{76}\text{Ge}$ , 36.5%  $^{74}\text{Ge}$ , 7.8%  $^{73}\text{Ge}$ , 27.5%  $^{72}\text{Ge}$ , and 20.5%  $^{70}\text{Ge}$ . Whereas, gallium in its naturally occurring form is composed of about 60.4%  $^{69}\text{Ga}$  and 39.6%  $^{71}\text{Ga}$ . Isotopically enriched forms of these materials include any composition in which the percentage of one isotope is increased in comparison to all other isotopes. Reducing the presence of other isotopes (i.e., increasing the isotopic purity of the material) will increase both the carrier mobilities and thermal conductivity of the resulting material. In general, it is desirable to use isotopically enriched material in which the other isotopes represent less than 6%, and more preferably, less than 2% of the composition. Thus, for example, isotopically enriched germanium for use in device fabrication might include more than 94%  $^{74}\text{Ge}$  and less than 6% of the other isotopes of Ge. Similarly, isotopically enriched gallium in GaAs that is used for device fabrication might include more than 94%  $^{69}\text{Ga}$  and less than 6% of  $^{71}\text{Ga}$ .

In the above examples, the isotope that was present in the naturally occurring semiconductor in the greatest proportions was selected as the dominant one in the enriched material. Note, however, that the objective is to reduce the presence all but one isotope.

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Therefore, any of the isotopes could be selected as the dominant isotope in the enriched material.

As with silicon, other isotopically enriched source material (i.e., powdered semiconductor material) can also be obtained from commercial suppliers who may use known techniques to produce the material. Once the isotopically enriched source material is obtained, devices may be fabricated from it by using any of the commonly known device fabrication technologies that are currently used to produce devices from the naturally occurring compositions of the semiconductor materials. In addition to the technologies already mentioned for silicon, examples of commonly used fabrication technologies for GaAs and germanium include molecular beam epitaxy (MBE) and organometallic chemical vapor deposition (OMCVD).

The devices which may be constructed using the isotopically enriched semiconductor materials include all of the devices currently being fabricated using the naturally occurring semiconductor materials. For example, GaAs has commonly been used to build lasers (i.e., opto-electronic devices) such as the double-heterostructure laser 30 shown in Fig. 3, MESFET's (metal-semiconductor field effect transistors), and schottky devices and germanium has been used alone to construct germanium a variety of junction devices, such as diode 40 in Fig. 4, and in combination with silicon to construct multilayer Si-Ge structures. Using isotopically enriched semiconductors in these applications will improve those aspects of device performance related to the higher carrier mobilities and the higher thermal conductivities of the enriched materials.

What is claimed is:

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CLAIMS

- 1           1. A semiconductor structure comprising:  
2           a single-crystal region composed of an  
3 isotopically enriched semiconductor material; and  
4           a semiconductor device formed in said isotopically  
5 enriched semiconductor region.
- 1           2. The semiconductor structure of claim 1 further  
2 comprising a substrate and wherein said isotopically  
3 enriched semiconductor region is formed on one surface of  
4 said substrate.
- 1           3. The semiconductor structure of claim 2 wherein  
2 said isotopically enriched semiconductor region is a  
3 layer formed on one surface of said substrate.
- 1           4. The semiconductor structure of claim 1 wherein  
2 said semiconductor material is silicon.
- 1           5. The semiconductor structure of claim 4 wherein  
2 said silicon region has a higher proportion of one of the  
3 isotopes of Si than is present in naturally occurring  
4 silicon.
- 1           6. The semiconductor structure of claim 5 wherein  
2 said silicon region has a higher proportion of the  $^{28}\text{Si}$   
3 isotope than is present in naturally occurring silicon.
- 1           7. The semiconductor structure of claim 6 wherein  
2 said silicon region is composed of at least 94 %  $^{28}\text{Si}$ .
- 1           8. The semiconductor structure of claim 7 wherein  
2 said silicon region is composed of at least 98 %  $^{28}\text{Si}$ .

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1           9. The semiconductor structure of claim 1 wherein  
2 said semiconductor material is germanium.

1           10. The semiconductor structure of claim 9  
2 wherein said germanium region has a higher proportion of  
3 one of the isotopes of Ge than is present in naturally  
4 occurring germanium.

1           11. The semiconductor structure of claim 10  
2 wherein said germanium region has a higher proportion of  
3 the  $^{74}\text{Ge}$  isotope than is present in naturally occurring  
4 germanium.

1           12. The semiconductor structure of claim 11  
2 wherein said germanium region is composed of at least 94  
3 %  $^{74}\text{Ge}$ .

1           13. The semiconductor structure of claim 12  
2 wherein said germanium region is composed of at least 98  
3 %  $^{74}\text{Ge}$ .

1           14. The semiconductor structure of claim 1  
2 wherein said semiconductor material is GaAs.

1           15. The semiconductor structure of claim 14  
2 wherein the gallium in said GaAs region has a higher  
3 proportion of one of the isotopes of Ga than is present  
4 in naturally occurring gallium.

1           16. The semiconductor structure of claim 15  
2 wherein the gallium in said GaAs region has a higher  
3 proportion of the  $^{69}\text{Ga}$  isotope than is present in  
4 naturally occurring germanium.

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1           17. The semiconductor structure of claim 16  
2 wherein the gallium in said GaAs region is composed of at  
3 least 94 % <sup>69</sup>Ga.

1           18. The semiconductor structure of claim 17  
2 wherein said gallium in said GaAs region is composed of  
3 at least 98 % <sup>69</sup>Ga.

1           19. The semiconductor structure of claim 4, 13,  
2 or 25 wherein the semiconductor device is an electronic  
3 device.

1           20. The semiconductor structure of claim 4, 13,  
2 or 25 wherein the semiconductor device is an electro-  
3 optical device.

1           21. The semiconductor structure of claim 4, 13,  
2 or 25 wherein the semiconductor device is an opto-  
3 electronic device.

1           22. The semiconductor structure of claim 1  
2 further comprising an integrated circuit fabricated in  
3 the isotopically enriched semiconductor region.

1           23. The semiconductor structure of claim 1  
2 wherein said isotopically enriched semiconductor region  
3 primarily serves to provide heat dissipation for said  
4 device.

1           24. A method for fabricating a semiconductor  
2 structure comprising:  
3           forming a single-crystal region composed of  
4 isotopically enriched semiconductor material; and  
5           fabricating a semiconductor device in said  
6 isotopically enriched region.



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1           25. The method of claim 24 wherein said  
2 isotopically enriched semiconductor region is formed on  
3 one surface of a substrate.

1           26. The method of claim 25 wherein said  
2 isotopically enriched semiconductor region is a layer  
3 formed on one surface of said substrate.

1           27. The method of claim 26 wherein said step of  
2 forming the isotopically enriched semiconductor region  
3 involves epitaxial deposition.

1           28. The method of claim 24 wherein said  
2 semiconductor material is silicon.

1           29. The method of claim 28 wherein said silicon  
2 region has a higher proportion of one of the isotopes of  
3 Si than is present in naturally occurring silicon.

1           30. The method of claim 29 wherein said silicon  
2 region is composed of at least 94 %  $^{28}\text{Si}$ .

1           31. The method of claim 30 wherein said silicon  
2 region is composed of at least 98 %  $^{28}\text{Si}$ .

1           32. The method of claim 24 wherein said  
2 semiconductor material is germanium.

1           33. The method of claim 32 wherein said germanium  
2 region has a higher proportion of one of the isotopes of  
3 Ge than is present in naturally occurring germanium.

1           34. The method of claim 33 wherein said germanium  
2 region is composed of at least 94 %  $^{74}\text{Ge}$ .

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1           35. The method of claim 34 wherein said germanium  
2 region is composed of at least 98 %  $^{74}\text{Ge}$ .

1           36. The method of claim 24 wherein said  
2 semiconductor material is GaAs.

1           37. The method of claim 36 wherein the gallium in  
2 said GaAs region has a higher proportion of one of the  
3 isotopes of Ga than is present in naturally occurring  
4 gallium.

1           38. The method of claim 37 wherein the gallium in  
2 said GaAs region is composed of at least 94 %  $^{69}\text{Ga}$ .

1           39. The method of claim 38 wherein said gallium  
2 in said GaAs region is composed of at least 98 %  $^{69}\text{Ga}$ .

1           40. A semiconductor structure comprising an  
2 isotopically enriched single-crystal silicon region, said  
3 enriched silicon region being composed of at least 94% of  
4 one of the isotopes of Si.

1           41. The semiconductor structure of claim 40  
2 further comprising a substrate and wherein said  
3 isotopically enriched silicon region is formed on one  
4 surface of said substrate.

1           42. The semiconductor structure of claim 40 or 41  
2 wherein said enriched silicon region is composed of at  
3 least 94% of  $^{28}\text{Si}$ .

1           43. The semiconductor structure of claim 42  
2 wherein the proportion of  $^{28}\text{Si}$  in the isotopically  
3 enriched region is no less than 98%.

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1           44. A semiconductor structure comprising a  
2 single-crystal GaAs region wherein the gallium in said  
3 GaAs region is composed of at least 94% of one of the  
4 isotopes of Ga.

1           45. The semiconductor structure of claim 44  
2 further comprising a substrate and wherein said  
3 isotopically enriched GaAs region is formed on one  
4 surface of said substrate.

1           46. The semiconductor structure of claim 44 or 45  
2 wherein the gallium in said isotopically enriched GaAs  
3 region is composed of at least 94% of  $^{69}\text{Ga}$ .

1           47. The semiconductor structure of claim 46  
2 wherein the proportion of  $^{69}\text{Ga}$  in the isotopically  
3 enriched region is no less than 98%.

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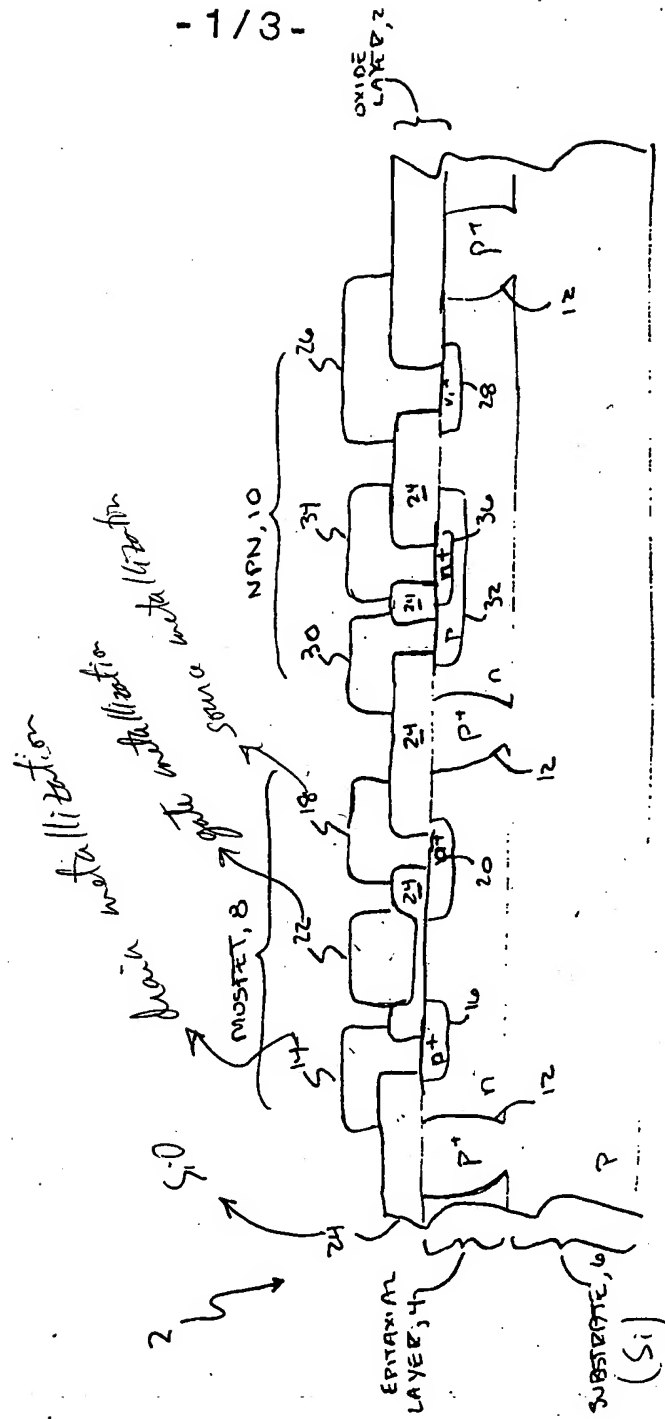


FIG. 1

- 2 / 3 -

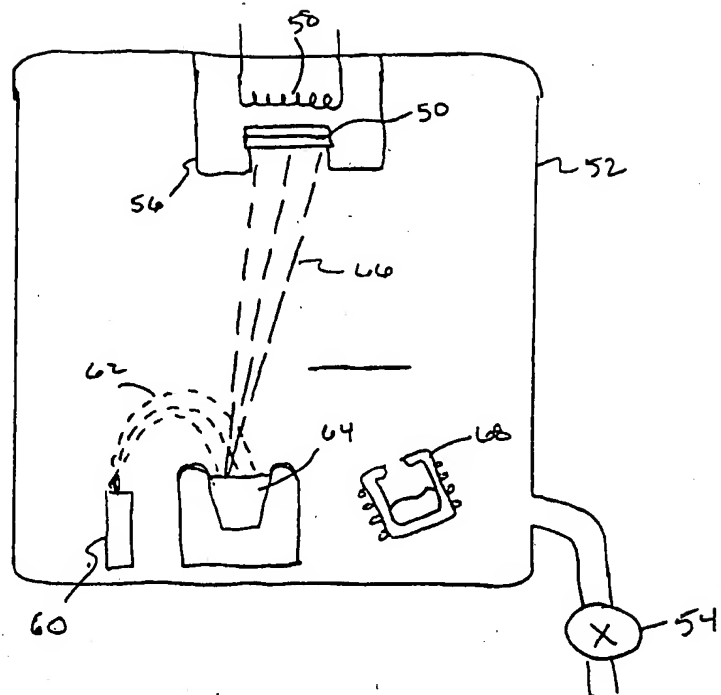


FIG. 2

- 3 / 3 -

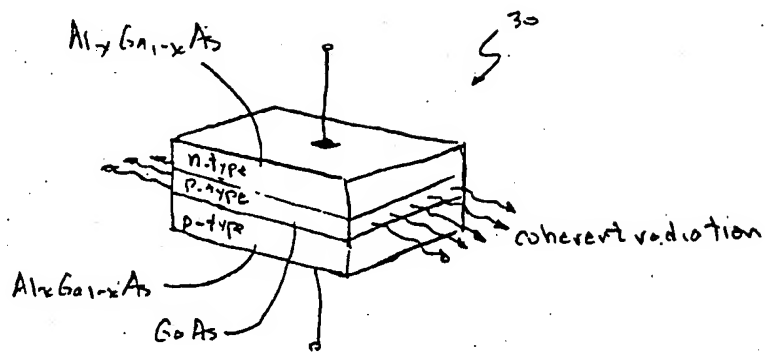


FIG. 3

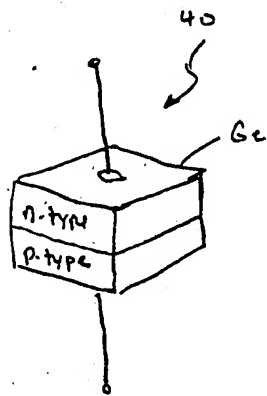


FIG. 4

# INTERNATIONAL SEARCH REPORT

International Application No. PCT/US91/06424

<b>I. CLASSIFICATION F SUBJECT MATTER</b> (if several classification symbols apply, indicate all) *		
According to International Patent Classification (IPC) or to both National Classification and IPC		
IPC(5): H01L 27/02, 21/26		
U.S. CL. 357/43,91; 437/173		
<b>II. FIELDS SEARCHED</b>		
Minimum Documentation Searched <sup>7</sup>		
Classification System	Classification Symbols	
U.S.	357/43,91; 437/173	
Documentation Searched other than Minimum Documentation to the extent that such Documents are included in the Fields Searched *		
<b>III. DOCUMENTS CONSIDERED TO BE RELEVANT *</b>		
Category *	Citation of Document, <sup>11</sup> with indication, where appropriate, of the relevant passages <sup>12</sup>	Relevant to Claim No. <sup>13</sup>
<u>Y</u> <u>A</u>	US, A, 4,721,684 (MUSUMECI) 26 JANUARY 1988 See whole document, especially Figure 1e.	1-5, 7, 8 22, 23 40-44 24 - 39
<u>Y</u> <u>A</u>	US, A, 4,628,341 (THOMAS) 09 DECEMBER 1986 See whole document, especially Figure 8.	1-5 22, 23 40, 41 24 - 39
<u>Y</u> <u>A</u>	JP, A, 63-269573 (NIPPON) 07 NOVEMBER 1988 See whole document, especially Figure 1.	1-5, 7, 8 22, 23 40 - 44 24- 39
<u>Y</u> <u>A</u>	JP, A, 61-274322 (MITSUBISHI) 04 DECEMBER 1986 See whole document, especially Figures 1 and 2.	1-5 22, 23 40, 41 24 - 39
<u>Y</u> <u>A</u>	JP, A, 55- 93256 (SONY) 15 JULY 1980, See Whole Document, especially Figure 3	1-5 22, 23 40, 41 24-39
<p>* Special categories of cited documents: <sup>10</sup></p> <p>"A" document defining the general state of the art which is not considered to be of particular relevance</p> <p>"E" earlier document but published on or after the international filing date</p> <p>"L" document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified)</p> <p>"O" document referring to an oral disclosure, use, exhibition or other means</p> <p>"P" document published prior to the international filing date but later than the priority date claimed</p> <p>"T" later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention</p> <p>"X" document of particular relevance: the claimed invention cannot be considered novel or cannot be considered to involve an inventive step</p> <p>"Y" document of particular relevance: the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art.</p> <p>"d" document member of the same patent family</p>		
<b>IV. CERTIFICATE N</b>		
Date of the Actual Completion of the International Search	Date of Mailing of this International Search Report	
14 JANUARY 1992	29 JAN 1992	
International Searching Authority	Signature of Authorized Official	
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